

SYNTHESIS AND APPLICATION OF MIXED SPINEL $\text{Mn}_{0.4}\text{Cd}_{0.6}\text{Cr}_2\text{S}_2\text{Se}_2$: STRUCTURAL, MAGNETIC, AND ELECTROCHEMICAL SENSING PROPERTIES

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ABSTRACT

Herein we report the solid synthesis, structural characterization, magnetic behavior, and electrochemical sensing properties of $\text{Mn}_{0.4}\text{Cd}_{0.6}\text{Cr}_2\text{S}_2\text{Se}_2$. Single-crystal X-ray diffraction analysis showed that $\text{Mn}_{0.4}\text{Cd}_{0.6}\text{Cr}_2\text{S}_2\text{Se}_2$ crystallizes in a spinel-type structure. Powder X-ray diffraction patterns and Rietveld refinement data revealed that this selenide phase is consisted of cubic $\text{Fd}\bar{3}\text{m}$ space group. Magnetic field cooling (MFC) measurements indicated an enhancement in ferromagnetic interactions relative to the ferrimagnetic compound $\text{Mn}_{0.4}\text{Cd}_{0.6}\text{Cr}_2\text{S}_4$, which can be attributed to the substitution of sulphur by selenium. The electrochemical response of modified glassy carbon electrodes with $\text{Mn}_{0.4}\text{Cd}_{0.6}\text{Cr}_2\text{S}_2\text{Se}_2$ was increased, the peak current is increased 4-fold, from 20.15 μA to 83.52 for GC, and GC- $\text{Mn}_{0.4}\text{Cd}_{0.6}\text{Cr}_2\text{S}_2\text{Se}_2$ respectively by differential pulse voltammetry, and thus it could be used to design an electrochemical sensor to quantify nitrocompounds, considered pollutants and toxic agents for humans, plants, and animals.

Keywords: Chalcospinel, electrochemical sensor, ferromagnetism, single crystal.

1. INTRODUCTION

The use of magnetic materials in electrochemical detection has a great potential for detecting and performing various analyses. This is due to the special properties of magnetic materials, which can enhance the efficacy and sensitivity of electrochemical sensors¹. These materials can be used to identify biological components, pollutants in the environment, and chemical species, among other analyses^{2,3}. Mainly oxides have been used as sensors for various species, such as, whether environmental pollutants⁴, drugs⁵, volatile organic compounds (VOCs)⁶ and nitro aromatic compounds (NACs)⁷⁻⁹, among others¹⁰⁻¹⁴, but it has been demonstrated in some cases that sulfides have richer redox reactions as well as higher conductivity, they have the ability to offer multiple reaction sites, extended mass and charge transport channels, and elevated surface-to-volume ratios^{12,15-17}. Furthermore, they outperform metal oxides in terms of mechanical, thermal, and electrical conductivities¹⁷⁻¹⁹.

These oxides materials correspond to spinels that are a large group of compounds having the general structure $(\text{A})[\text{B}_2]\text{X}_4$, with the same crystal structure as the natural mineral MgAl_2O_4 , $\text{Fd}\bar{3}\text{m}$ (227), where the square brackets indicate the octahedral sites and the parenthesis indicate the tetrahedral ones (e.g., examples ACr_2X_4 , A = Mn, Fe, Co, Cu, Zn, and often X are O^{2-} , S^{2-} , or $\text{Te}^{2-20-28}$). Over time, more than 200 distinct spinels with the general formula AB_2X_4 have been discovered or produced in single- or polycrystalline form and they became significant early on as ferromagnetic or ferrimagnetic insulators with high ordering temperatures and rather high saturation magnetization²¹. Has been found, electrical conductivities of chalcospinel (AB_2X_4 with X = S^{2-} or Se^{2-}), mechanical and thermal stabilities are superior to those of their corresponding metal oxides¹⁷ and modeling the magnetic properties of these spinels is achieved by replacing the cations and chalcogen that make them up. There are various synthesis methods to obtain chalcospinel, but the conventional ceramic method can be advantageous owing to its simple operation, low cost, high yield, and bulk production^{29,30}. Allowing its direct use in different applications such as the substances electrochemical quantification.

On the other hand, Nitrophenol (NP) derivatives are compounds extensively used in industry, and they are considered toxic agents for humans, plants, and animals³¹⁻³³. 4-Nitrophenol (4-NP) is the most common and abundant of nitrophenol compounds, thus, it is important to develop strategies to quantify this analyte³⁴. There are several methodologies for 4-NP quantification, Liquid Chromatography, UV-vis Spectrophotometry, and Chemiluminescence, with good performances, but with a high cost, limiting their use³⁵. As an alternative, the electrochemical methods to quantify 4-NP have the advantage of lower cost, easier operation, and short-time response, compared with the above-mentioned, and this is the reason the developing of sensors with new materials that improve electrochemical response is an interesting field to investigate^{36,37}.

In this work, we present the synthesis, structural, and magnetic characterization of $\text{Mn}_{0.4}\text{Cd}_{0.6}\text{Cr}_2\text{S}_2\text{Se}_2$, as well as the study on the reduction of 4-Nitrophenol (4-NP) for promising candidate for the design of electrochemical sensors for the quantification of nitrocompounds using magnetic mixed chalcospinel.

2. EXPERIMENTAL SECTION

2.1 Materials

All the reagents used for the synthesis of these compounds were obtained commercially from Sigma-Aldrich, St. Louis, USA, and used without further purification.

2.2 Synthesis

Polycrystalline $\text{Mn}_{0.4}\text{Cd}_{0.6}\text{Cr}_2\text{S}_2\text{Se}_2$ compounds were prepared using ceramic method at high temperature by directly combining high-purity elemental powders (99.99%, Aldrich) in stoichiometric amounts. All manipulations were carried out under an argon atmosphere. The reaction mixtures were sealed in evacuated quartz ampoules and placed in a programmable furnace. The ampoules were then slowly heated from room temperature to 800 °C at 60 °C/h and held at the maximum temperature for 8 days, then ampoules were cooled to room temperature to be opened and ground the sample, resealed, and heated again to 800 °C at 60 °C/h for 8 days more. Finally, the ampoules were cooled to room temperature, they were opened and ground for further characterization.

2.3 Characterization

2.3.1 Powder X-ray diffraction measurements

Powder X-ray diffraction (PXRD) patterns were collected at room temperature on Bruker D8 Advance diffractometer equipped with a $\text{Cu K}\alpha$ radiation source, in a range the range $5^\circ < 2\theta < 80$. The XRD patterns were indexed using the CHEKCELL software³⁸. The collected data were analyzed by Fullprof Rietveld refinement software³⁹. A LaB_6 powder standard was used to determine the instrumental profile.

2.3.2 Single Crystal X-ray Diffraction

Single-crystal X-ray diffraction (XRD) data for $\text{Mn}_{0.4}\text{Cd}_{0.6}\text{Cr}_2\text{S}_2\text{Se}_2$ were recorded at RT with a D8 Venture Bruker AXS diffractometer equipped with $\text{MoK}\alpha$ radiation ($\lambda = 0.71073 \text{ \AA}$). Data were accumulated using the Bruker SMART software package⁴⁰. Structural refinement was performed using the SHELXL⁴¹ and Olex2⁴² programs.

2.3.3 Magnetic Measurements

Magnetic measurements were performed on pelletized powder samples using a physical property measurement system (PPMS-9T), manufactured by Quantum Design, in a temperature range from 5 K to 300 K. Continuous susceptibility $\chi_{dc}(T) = M(T)/H$ measurements were obtained for magnetization field cooling (MFC) with applied magnetic field $H = 0.05$ T.

2.3.4 Electrochemical Measurements

Electrochemical measurements were performed with a potentiostat CHI650E from CH Instruments, USA.

A three-electrode arrangement cell with PBSx10 at pH 7.44 was used. Glassy carbon electrode, GC, (GC CH Instruments CHI104) as a working electrode, Ag/AgCl (CHI111 3 M KCl) as a reference electrode, and Pt wire (CHI115) as a counter electrode. The cell was purged with Nitrogen for 15 minutes before the electrochemical measurements, and the pressure was maintained during all experiments.

Electrode modification

Glassy carbon electrodes were previously polished with 0.3 and 0.05 μm alumina and then washed with deionized water and dried at room temperature to be modified using a drop cast technique with an aliquot of 20 μL of 3 mg/mL of $\text{Mn}_{0.4}\text{Cd}_{0.6}\text{Cr}_2\text{S}_2\text{Se}_2$ dispersion in isopropanol onto the electrode surface. The

modified electrodes were allowed to air dry at room temperature to obtain GC- $\text{Mn}_{0.4}\text{Cd}_{0.6}\text{Cr}_2\text{S}_2\text{Se}_2$ electrodes.

4-Nitrophenol quantification

Using the electrochemical cell conditions mentioned before, the addition of 4-nitrophenol, 4-NP, was performed to evaluate the GC- $\text{Mn}_{0.4}\text{Cd}_{0.6}\text{Cr}_2\text{S}_2\text{Se}_2$ electrode used as a possible sensor for nitro compounds.

3. RESULTS AND DISCUSSION

3.1. X-ray powder diffraction and compositional characterization

The crystal structure of $\text{Mn}_{0.4}\text{Cd}_{0.6}\text{Cr}_2\text{S}_2\text{Se}_2$ was resolved by single-crystal X-ray diffraction in the cubic $\text{Fd}\bar{3}\text{m}$ space group. The least-squares refinement of the occupation factors and the displacement parameters converged to a model in which the tetrahedral (A) positions were occupied by Cd and Mn (8a Wyckoff sites) and the octahedral [B] positions (16d Wyckoff sites) were occupied by Cr. The S and Se atoms occupy the 32e (u,u,u) Wyckoff sites in a cubic closed-packed array. This model was based on linear functions constraining the Mn, Cd, and Cr atoms at the 8a and 16d sites. The sum of the site occupation factor (SOF) was forced to equal 1 (fully occupied for $8a = (1-x)\text{Mn} + x\text{Cd}$). Furthermore, for the chalcogenide site, the sum of the SOF was forced to equal 1 (fully occupied) for $32e = (4-y)\text{S} + y\text{Se}$, and the Mn/Cd and S/Se atoms were also constrained to have equivalent atomic displacement parameters (ADPs). The detailed crystallographic data and refinement results are summarized in Tables 1.

Table 1. Crystallographic data and structure refinement details CSD-2368795.

Empirical formula	$\text{Mn}_{0.40}\text{Cd}_{0.60}\text{Cr}_2\text{S}_2\text{Se}_2$
Formula weight	413.06
Temperature/K	300
Crystal system	cubic
Space group	$\text{Fd}\bar{3}\text{m}$
$a/\text{\AA}$	10.4157(5)
$b/\text{\AA}$	10.4157(5)
$c/\text{\AA}$	10.4157(5)
$\alpha/^\circ$	90
$\beta/^\circ$	90
$\gamma/^\circ$	90
Volume/ \AA^3	1129.97(16)
Z	8
$\rho_{\text{calc}}/\text{g/cm}^3$	4.916
μ/mm^{-1}	20.434
$F(000)$	1503
Crystal size/ mm^3	$0.05 \times 0.045 \times 0.022$
Radiation	$\text{MoK}\alpha$ ($\lambda = 0.71073$)
2θ range for data collection/ $^\circ$	6.776 to 60.65
Index ranges	$-14 \leq h \leq 14, -14 \leq k \leq 14, -14 \leq l \leq 14$
Reflections collected	5229
Independent reflections	108 [$R_{\text{int}} = 0.0954, R_{\text{sigma}} = 0.0228$]
Data/restraints/parameters	108/0/8
Goodness-of-fit on F^2	1.122
Final R indexes [$l > 2\sigma(l)$]	$R_1 = 0.0259, wR_2 = 0.0671$
Final R indexes [all data]	$R_1 = 0.0319, wR_2 = 0.0721$
Largest diff. peak/hole / $e \text{\AA}^{-3}$	0.879/-0.657

The Cr-X (X = Se/S) bond distance in $\text{Mn}_{0.40}\text{Cd}_{0.60}\text{Cr}_2\text{S}_2\text{Se}_2$ was 2.4824(5) \AA which is comparable with other Cr-spinels²⁹. Cd/Mn-X (X = Se/S) distance in the sample was 2.4771(10) \AA higher than informed for Mn-S distance 2.387 \AA in MnCr_2S_4 ⁴³. This is due to the presence of Cd in the Mn site and S in the Se site, both of which have a larger ionic radius, resulting in an increase in the unit cell size. The bond angles of X-Cd/Mn-X is 109.5, bond angle of X-Cr-X (X = Se/S) (180.0, 96.07 and 83.93) in the sample are very similar to the bond angles in $\text{CuCr}_{1.1}\text{Sn}_{0.9}\text{S}_{2.3}\text{Se}_{1.7}$ ²⁹ (180.0, 92.28 and 87.72). The degrees of distortion of the

tetrahedra or octahedra were evaluated using the distortion indices (DI) defined by Baur and Wildnerwil⁴⁴⁻⁴⁶. The $\text{Mn}_{0.40}\text{Cd}_{0.60}\text{Cr}_2\text{S}_2\text{Se}_2$ compound have three-atom centered polyhedral structure: (A) X_4 (tetrahedron), [B] X_6 (octahedron) and X[B₃A] with A = Mn/Cd, B = Cr and X = S/Se. The degree of distortion in the polyhedral can be measured using the edge length distortion (ELD) indices. The X[B₃A] tetrahedron is the most distorted polyhedron (~8% distortion from an ideal tetrahedron)⁴⁷. These values compare very well with those found for other

Cr-spinels²⁹. The (A)X₄ tetrahedra were ideal, with tetrahedral angles 109.5°. The [B]X₆ was slightly distorted indicating a near-ideal octahedral structure.

The XRD pattern was fully indexed in the space group $Fd\bar{3}m$ (N° 227), with the exception of two very weak impurity peaks, within the detection limits of the technique (Figure 1). These peaks correspond to the MnCr₂S₄ phase present less than 1% as reported by the Rietveld analysis (Figure 2). Lattice parameters of powder sample calculated by Rietveld refinement are shown in Table 2. The experimental XRD patterns were compared with the simulated XRD patterns derived from the single-crystal XRD data. The shape and intensity of the XRD peaks indicates the high crystallinity, and the observed interlayer spacing was in good agreement with the calculated interplanar spacing *d*.

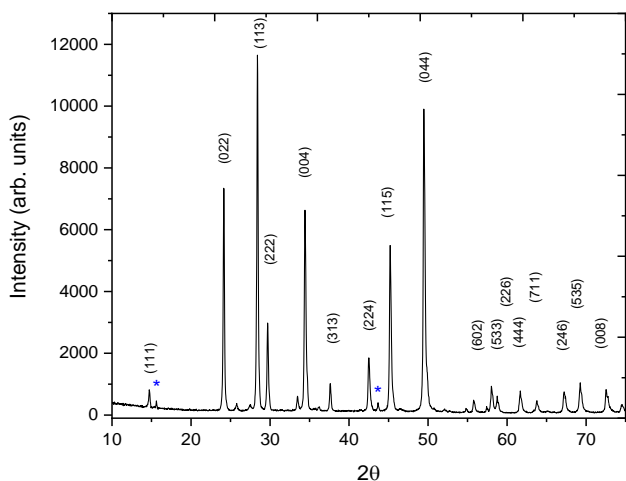


Figure 1. X-ray powder diffraction patterns for Mn_{0.4}Cd_{0.6}Cr₂S₂Se₂. The blue asterisks indicate reflections associated with a MnCr₂S₄ impurity.

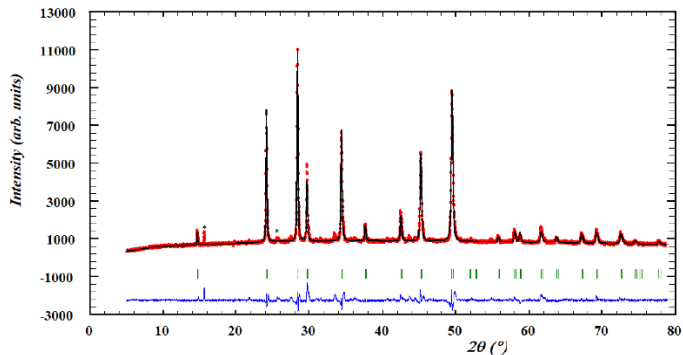


Figure 2. Rietveld refinement of powder diffraction pattern of Mn_{0.4}Cd_{0.6}Cr₂S₂Se₂. Black solid line is calculated pattern, red dashed line is experimental data, the green ticks are Bragg positions, and the bottom blue line is the difference of experimental and calculated patterns.

Table 2. Results of Rietveld refinement and agreement factors from powder diffraction data for Mn_{0.4}Cd_{0.6}Cr₂S₂Se₂.

Empirical formula	Mn _{0.4} Cd _{0.6} Cr ₂ S ₂ Se ₂
Crystal system	Cubic
Space group	$Fd\bar{3}m$
<i>a</i> /Å	10.41648(7)
<i>V</i> /Å ³	1130.22(9)
<i>d</i> /g cm ⁻³	4.883
Agreement factors	
<i>R_p</i>	5.1
<i>R_{wp}</i>	7.1
χ^2	5.5

Lattice parameter *a* calculated (10.416 Å) is greater than the system Mn_{0.4}Cd_{0.6}Cr₂S₄ (10.192 Å) obtained in 2010 by Barahona and coworkers³⁰. This fact could be related to the difference of ion size radii of Se²⁻ (1.98 Å) and S²⁻ ions (1.84 Å), which causes an increase in the network parameters⁴⁸. In addition, this value is close to that reported for the spinel of the MnCr₂S₂Se₂ type, which is 10.315 Å⁴⁹. Is important to note that the larger size of Cd²⁺ (0.92 Å) compared to Mn²⁺ (0.80 Å) also affect the cell parameter. Therefore, there is a simultaneous effect of both the chalcogen (Se²⁻) and the ion-metal (Cd²⁺). The size of unit cell will increase consequently as substitution of Selenium in the crystallographic sites of Sulphur. The average grain size, which is determined from grazing incidence XRD using the Debye–Scherrer formula⁵⁰ with the full width at half maximum of the (022) diffraction peak, is ~45.5 nm.

3.2 Magnetic Measurements

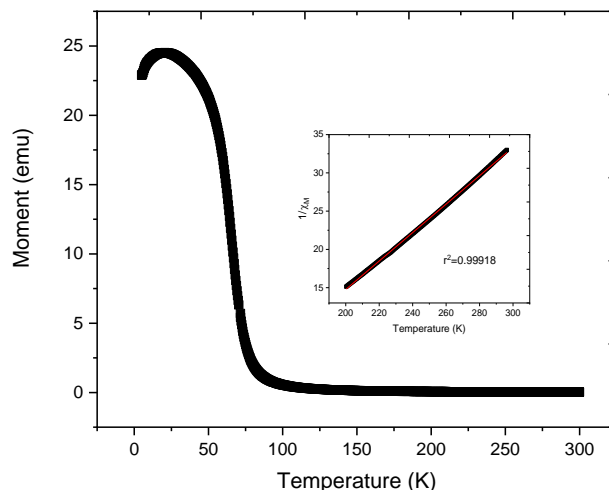


Figure 3. MFC magnetization for Mn_{0.4}Cd_{0.6}Cr₂S₂Se₂

The Figure 3 shows MFC measurements on Mn_{0.4}Cd_{0.6}Cr₂S₂Se₂, the paramagnetic regime was identified well above the magnetic transition temperature at 500 Oe (inset Figure 3). The inverse susceptibility was fitted by a classical Curie-Weiss relation $c = C/(T-q)$ in the range $200 \leq T \leq 300$ K. The effective moment m_{eff} obtained was 6.56 MB, this value agrees with what can be expected for Mn²⁺ and Cr³⁺ free-spin values. The Curie-Weiss temperature was +120 K, this value that indicates a strong ferromagnetic component in the ferrimagnetic behavior of the material.

In our work, we define the paramagnetic to ferrimagnetic transition *T_c* at the temperature where a change in the slope in MFC of the magnetic measurements occurs, i.e. *T_c* = 66.1 (±1) K, while *T_{max}* ~20.6 (±1)K. *T_{max}* corresponds to a reorientation of the manganese spins in the presence of the internal field created by the ferromagnetic chromium sublattice³⁰.

The transition temperature *T_c* = 66.1 K of Mn_{0.4}Cd_{0.6}Cr₂S₂Se₂ should be compared with that obtained in polycrystalline Mn_{0.4}Cd_{0.6}Cr₂S₄ (*T_c* = 79 K). The decrease in the transition temperature from paramagnetic to ferrimagnetic can be associated with the replacement of sulphur with selenium in the structure, similarly to MnCr₂S_{4-x}Se_x where an increase in the amount of selenium causes a decrease in the *T_c* value from 72 to 63 K for *x* = 0.1 to 1⁴⁹. With respect to the ferromagnetic interaction parameter, the value +120 K higher compared to +30 K for MnCr₂S₄ and +91 ± 5K for Mn_{0.4}Cd_{0.6}Cr₂S₄, this can be explained with the increase in the ferromagnetic interactions of the chromium ions due to the replacement of sulphur by selenium, which has a larger size³⁰.

The magnetic parameters obtained for Mn_{0.4}Cd_{0.6}Cr₂S₂Se₂ indicate a stronger ferromagnetism when sulphur is replaced by selenium, this would be due to the fact that the antiferromagnetic interactions between the manganese and chromium sublattices are weakened by the increase in the separation of the magnetic ions, product of the presence of selenium, which is larger than sulphur, leading to well-aligned spins pointing in the same direction⁴⁸. The latter would favour the ferromagnetic interactions of the chromium moments and, on the other hand, the internal field due to the Cr lattice imposes a predominant orientation in the direction of the applied field.

3.3 Cyclic voltammetry

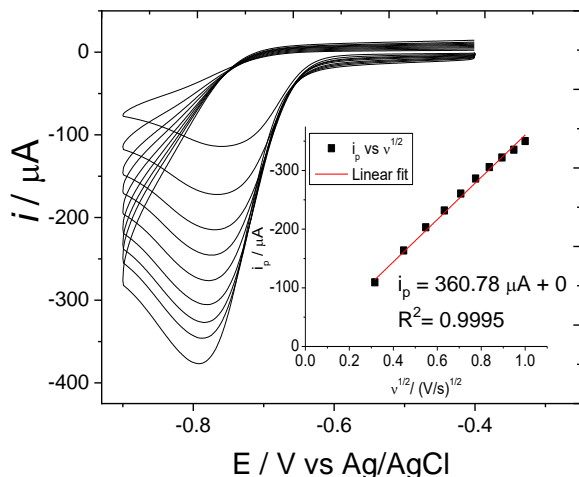


Figure 5. Cyclic voltammetry of 4NP 1 mM in PBS 10X at pH 7.44 at different sweep rates.

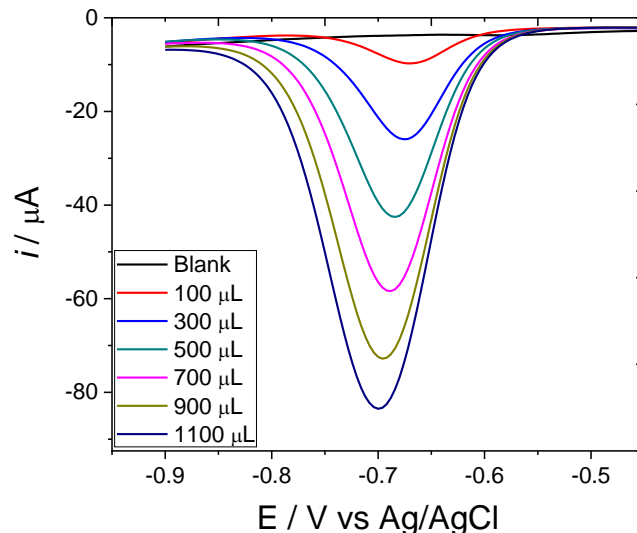


Figure 6. Extended cyclic voltammograms of 4NP 1 mM in PBS 10X at pH 7.44 at 0.1 Vs⁻¹.

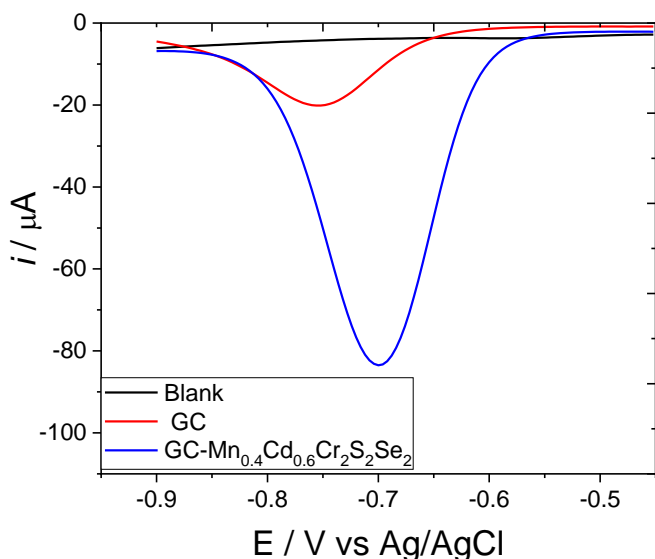


Figure 7. Differential pulse voltammograms of 4NP 1 mM in PBS 10X at pH 7.44

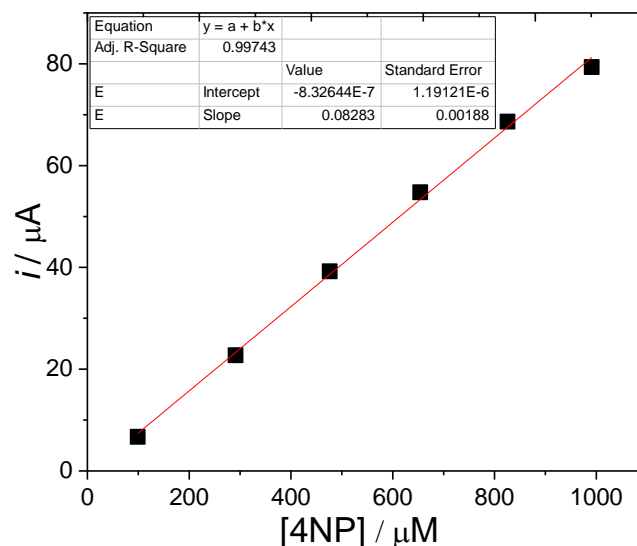


Figure 8. (up) Differential pulse voltammograms at different aliquots of 4NP 0.01 M in PBS 10X at pH 7.44 using GC-Mn_{0.4}Cd_{0.6}Cr₂S₂Se₂ and (down) its Calibration curve of DPV.

Figure 5 shows cyclic voltammograms from 0.1 - 1 Vs⁻¹ of 4NP 1 mM in PBS for the irreversible reduction process of the -NO₂ group, around at -0.79 V to -NHOH formation is attributed, i_p vs $v^{1/2}$ was plotted, and the linear tendency indicates the process is diffusion controlled. Subsequently, an extended cyclic voltammogram was performed, Figure 6, and a reversible peak was observed at 0.1 V, corresponding to -NHOH/-NO redox pair, which was isolated in the insert of Figure 6, with a negligible shift in $E_{1/2}$ values (0.089 V, and 0.083 V for GC and GC-Mn_{0.4}Cd_{0.6}Cr₂S₂Se₂ respectively). A DPV for comparison between GC and GC-Mn_{0.4}Cd_{0.6}Cr₂S₂Se₂ electrodes was made, Figure 8, with a slight shift from -0.753 and -0.700 V peak potential, E_p , respectively was observed. Also, the peak current is increased 4-fold, from 20.15 μA to 83.52 for GC, and GC-Mn_{0.4}Cd_{0.6}Cr₂S₂Se₂ correspondingly. These results indicate that Mn_{0.4}Cd_{0.6}Cr₂S₂Se₂ is a suitable material to modify electrodes for quantification purposes, thus, the response for -NO₂ reduction by DPV technique was followed, to an increasing concentration of 4-NP, shows an increase in the peak current, Figure 8 left. A calibration curve to assess a possible use as a sensor for this sample was made, and, subsequently, a peak current vs 4-NP concentration was plotted, Figure 8 right, and a good linear tendency was observed in the range of concentration used (99 μM to 0.99 mM) with $R^2 = 0.997$. All these results show that Mn_{0.4}Cd_{0.6}Cr₂S₂Se₂ is suitable for develops a sensor for 4-NP detection, this opens possibilities for the use of this material to study many other nitrocompounds, and also test other spinels for this purpose.

CONCLUSIONS

In summary polycrystalline $\text{Mn}_{0.4}\text{Cd}_{0.6}\text{Cr}_2\text{S}_2\text{Se}_2$ were successfully obtained using ceramic method at high temperature, single crystal was obtained as by-product of synthesis and this structure were confirmed by single crystal x-ray diffraction. MFC measurements on $\text{Mn}_{0.4}\text{Cd}_{0.6}\text{Cr}_2\text{S}_2\text{Se}_2$ indicates an increase of the ferromagnetic interaction with respect to the ferrimagnetic compound $\text{Mn}_{0.4}\text{Cd}_{0.6}\text{Cr}_2\text{S}_4$, when sulphur is replaced by selenium, which presents a larger size. The replacement of the anion causes the antiferromagnetic alignment between the Mn and Cr spins to decrease due to the separation of the ion sublattices. In addition, the sublattice separation causes the chromium ions to impose a predominant preferential orientation in the direction of the applied field leading to an overall ferromagnetic alignment. As we can see, using $\text{Mn}_{0.4}\text{Cd}_{0.6}\text{Cr}_2\text{S}_2\text{Se}_2$ increases the electrochemical response in modified glassy carbon electrodes, and thus it could be used to design an electrochemical sensor to quantify nitro-compounds or explore with other analytes quantification. The use of this material shows a promissory result to be considered in development of sensor of 4-NP or other environmental pollutants.

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